



The Effect of Selenium Source Temperature on the Structural Properties of Sb₂Se₃ Thin Films

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Opinion

Sb₂Se₂ is one of the best absorbent materials for next generation thin film solar cells and has excellent photovoltaic performance. Antimony selenide (Sb₂Se₂) is a promising alternative absorber material compared to the conventional thin film solar cells. Due to its outstanding properties, such as simple crystal structure, high absorption coefficient (>10⁵sm⁻¹), perfect optical band gap (1.1-1.3eV) and significant carrier mobility (~10sm²V⁻¹s⁻¹), Sb₂Se₂ has been regarded as one of the most attractive absorber candidates for the next-generation thin film solar cells. It should be noted that in the process of obtaining Sb₂Se₃ films by physical methods, a significant loss of Se occurs due to the process of decomposition of the films into Sb, Se, and SbSe during their synthesis. This leads to the formation of Se vacancies, which in turn increases the density of recombination centers in films [1]. This phenomenon negatively affects the optical and electrophysical properties of films and solar cells based on them. As being one of the most competitive absorber candidates for the next-generation thin film photovoltaic, Sb₂Se₃ has attracted much attention and so various deposition techniques; thus, thermal evaporation, Vapor Transport Deposition (VTD), Close-Spaced Sublimation (CSS) and sputtering [2] have been thoroughly studied to increase the PCE of the devices. The morphology and electrical properties of the films are found to be strongly dependent on the selenium source temperature, with higher temperatures resulting in larger crystalline grains and higher conductivity. These results suggest that the Chemical Molecular Beam Deposition (CMBD) method can be used to produce high-quality Sb₂Se₂ films with tunable properties for use in solar cell applications.

Experiment

Installed technological mode optimal cultivation quality Sb₂Se₂ films on surfaces glass (SLG-soda-lime glass) by the method chemical molecular beam deposition. The process of receiving Sb₂Se₃ films by the CMBD method was as follows: as original material used granules binary Sb₂Se₃ compound and Se semiconductor element purity (99.999%), which placed in different containers: Sb₂Se₂ into one and Se into the other. Further the system was brought to working condition and purged hydrogen for the purpose removal of atmospheric polluting gases. Dimensions samples -2.0×2.0cm². For receiving Sb₂Se₃ film enriched selenium and stoichiometric composition, it was changed partial pressure Se in the steam phase in progress their growth. In this case temperature substrates were 500 °C, temperature sources elements varied in the ranges: 350 °C÷430 °C for Se and 700 °C for Sb₂Se₂, the rate growth amounted to 0.1÷1Å/sec at stream hydrogen WH₂=20cm³/min. crystalline structure and phase compound materials explored method diffraction x-ray rays using-diffractometer. "Analytical Empyrean" on radiation CuK $\alpha(\lambda=1.5418\text{\AA})$ with a 2 θ measurement in the range from 20° to 80° in 0.01° increments. Analysis phase composition produced using-Joint Committee on Powder Diffraction Standard (JCPDS) bases. Dark CVCs of Sb₂Se₂ films were measured using a Keithley 2420 Source Mete.

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Result

Structural properties

Crystalline phases and crystalline structure of Sb₂Se₃ films were obtained at different selenium source temperatures. It can be seen from the X-ray patterns, that the total intensity of the (211), (221) peaks will sharply increase, while the (020), (230) and (240) peaks decrease at a selenium source temperature of 370 °C. Further increase in temperature to T_{se} =430 °C leads to a decrease in peaks

(221), (211) and weak peaks (020), (120), (230) and (240) increase significantly. Additionally, for 2θ =29.66° for peak (101) low reflex detected intensity, indicating the formation of the Se phase (Figure 1). To quantitatively study the orientation of Sb₂Se₃ films, the Texture Coefficients (TC) of diffraction peaks were calculated based on the following equation.

$$T_{c} = \frac{I_{(hkl)}}{I_{0(hkl)}} / \left(\frac{1}{N} \sum_{i=1}^{N} \frac{I_{(h_{i}k_{i}l_{i})}}{I_{0(h_{i}k_{i}l_{i})}}\right)$$
(1)



Figure 1: X-ray of Sb₂Se₃ films at different selenium source temperatures.

where I(hkl) and I_0 (hkl)-are the intensities of the diffraction peaks of the planes (hkl) on the measured and standard X-ray diffraction patterns of Sb_2Se_3 (JCPDS 15-0861), respectively. The large value of the TC of the diffraction peak indicates the predominant orientation in this direction. The TC values of the planes (hk0) of our samples tend to decrease at a selenium source temperature of 370 °C, and then begin to increase with a further increase in the selenium source temperature.

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