

Densification Kinetics of Titanium Carbide-Based Cermet During the Pressure Sintering

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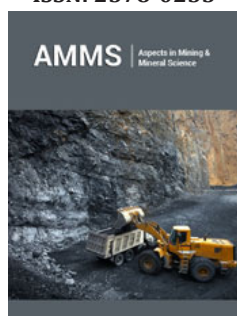
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Opinion

Titanium carbide-based cermets with the addition of other carbides and with metal binders of nickel, iron and their alloys are used in mechanical engineering technology and instrument-making industry as wear-resistant and corrosion-resistant materials and coatings. The goal remains to increase the mechanical and functional properties of cermets, and for this research is being conducted aimed at optimizing the compositions and improving the technology for their manufacture. This report presents the results of a study of nonisothermal pressure sintering a powder mixture containing, in addition to titanium carbide TiC, mass fractions of carbides: 4% VC, 6% Mo₂C, 6% NbC, and 20% NiCr. Pressure sintering of powder mixtures was carried out on an induction heating unit made on the basis of a hydraulic press and a thyristor converter in an atmosphere formed as a result of the interaction of a graphite die with air. After preliminary slow heating, a sintering regime was established with a controlled heating rate of 15K/min under a constant pressure of 20MPa. The obtained experimental data on the current change in relative density and temperature in the process of cermet sintering showed a regular acceleration of its densification with increasing temperature.

The processing of the experimental data was performed within the framework of the theory of bulk viscous flow of the porous body using the developed computer programs. The exact data on the densification rate was determined by the approximation method of the compaction curve by Lagrange polynomials, the differentiation of which yielded a derivative, that is, the densification rate, at each experimental point of the specified curve. A typical dependence of the relative density and temperature on the sintering time of the sample is presented below in the (Figure 1). The studied cermet is considered as a composite formed by a matrix having a theoretical material density and corresponding to a maximum relative density of 1 and pores whose density is equal to 0. Quantitative data on the relative density ρ and the densification rate enable the calculation of the current change in root-mean-square stress σ_m and strain rate in the matrix forming a porous composite.

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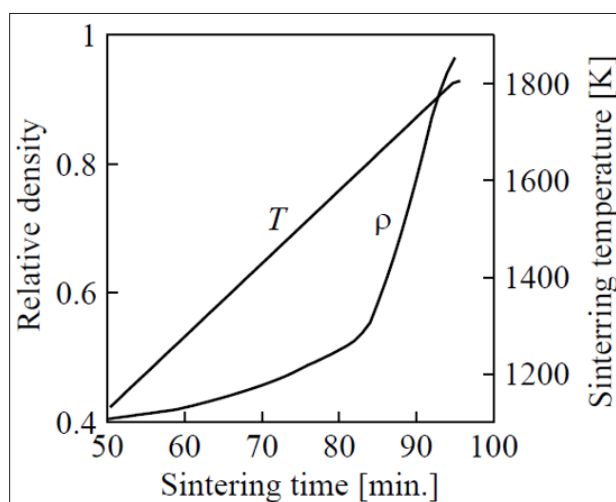


Figure 1: The cermet relative density ρ and temperature T versus sintering time t .

The quantities above are expressed by the formulae as follow:

$$\dot{\varepsilon}_m = \frac{1}{\rho} \frac{d\rho}{dt} \rho^{(2.5-2\rho)/(2\rho)} \sqrt{\frac{2-\rho^{2/\rho}}{2(1-\rho^{2/\rho})}} \quad \tau_m = \sigma_1 \sqrt{\frac{2(1-\rho^{2/\rho})}{\rho^{2.5/\rho}(2-\rho^{2/\rho})}}$$

Here σ_1 is the axial pressure and t is time.

The results of the calculation are showed in Figure 2. The root-mean-square stresses gradually decreases during the cermet densification from the initial value of 327.6MPa at a temperature 825 °C to 101.1MPa at 1298 °C and in the range of temperatures corresponding to sintering in the presence of the liquid phase, goes to zero. The root-mean-square rate gradually increases with the increase in temperature during the solid phase sintering

from $4.42 \cdot 10^{-5} \text{s}^{-1}$ at 825 °C to $1.43 \cdot 10^{-4} \text{s}^{-1}$ at 1298 °C. During the transition to the stage of liquid-phase sintering, an increase in the densification rate to $2.6 \cdot 10^{-4} \text{s}^{-1}$ at 1393 °C is replaced by a sharp decrease in it as the porous body densifies. The transition to the liquid-phase sintering stage is particularly clear on plots for logarithm of the relative density function rate [1] versus the reciprocal thermodynamic temperature (Figure 3), from which the activation energy of the matrix viscous flow was determined [1]. It is established that solid-phase pressure sintering is controlled by the mechanism of nonlinear creep ($n \gg 1.5$) with activation energy, which depends on the stress in the material, and liquid-phase sintering mechanism by virtually linear creep ($n=1$) with activation energy of 542kJ/mol.

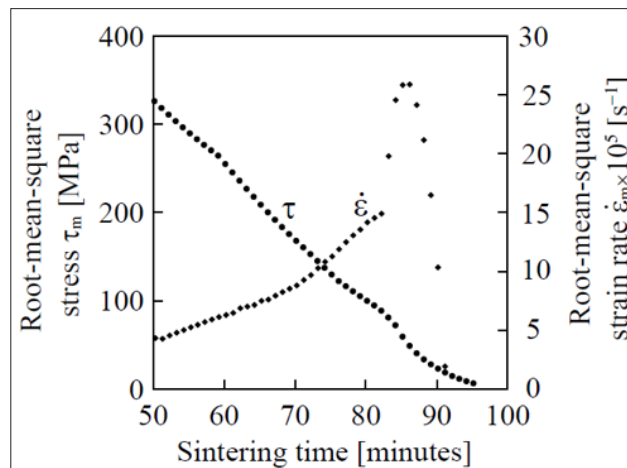


Figure 2: Root-mean-square stress τ_m and strain rate $\dot{\varepsilon}_m$ in the matrix.

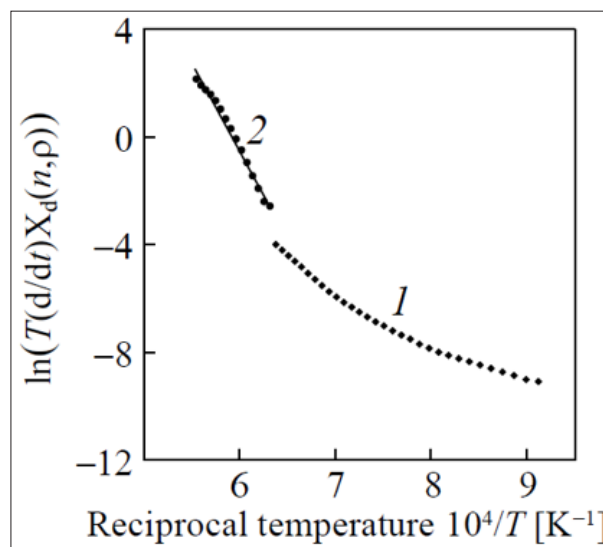


Figure 3: Logarithm of the relative density function rate $\ln(T(d/dt)X_d(n, \rho))$ vs. reciprocal temperature T^{-1} for solid-phase (1) and liquid-phase (2) pressure sintering.

References

1. Kovalchenko MS (2016) Rheology and kinetics of pressure sintering. Materials Science Forum 835: 76-105.

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